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Evaluation of the forming quality of Inconel 625 thin-walled parts manufactured via cold metal transfer additive manufacturing

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ABSTRACT

Introduction: Wire arc additive manufacturing (WAAM) is a metal additive manufacturing technique that uses an arc source to melt metallic wires, depositing molten metal layer by layer to form parts. Controlling the forming quality of parts in the WAAM process poses significant challenges. This study evaluated the shape quality of thin walls produced via WAAM using Inconel 625 alloy. Additionally, the impacts of the welding speed (v) and linear heat input (LHI) on the geometric quality of the fabricated components are investigated.

Methods: A cold metal transfer (CMT)-WAAM system was employed to construct thin-walled samples on low-carbon substrates. Three samples were fabricated at three different welding speeds (*v* = 35, 50, and 65 cm/min), while the other parameters remained constant. The samples were scanned via a Kreon Zypher II scanner, and their geometric properties, including average layer height (*ALH*), total wall width (*TW*), effective wall width (*EW*), and material deposition efficiency (*DE*), were measured via Geomagic Design X and AutoCAD software. Surface roughness parameters (e.g., *S^z* , *Sa*, and *Sq*) were assessed via Omnisurf 3D software.

Results and Discussion: Increasing the welding speed from 35 to 65 cm/min led to reductions in all measured characteristics - *ALH*, *TW*, *EW*, and *DE*. For example, *ALH* decreases from 2.63 mm to 1.87 mm, *TW* decreases from 9.39 mm to 6.83 mm, and *EW* decreases from 5.90 mm to 4.30 mm. An increase in the *LHI* from 19.08 to 35.22 J/mm tends to inversely affect these geometric characteristics. Additionally, as *v* or *LHI* increases, *S^z* , *Sa*, and *S^q* initially decrease to a certain level before rising again.

Conclusions: The results obtained from this study offer valuable insights into the relationship between processing and forming quality in the CMT-WAAM process for Inconel 625 thin-wall components. These insights provide a foundation for selecting optimal process parameters and providing informed recommendations for the CMT-WAAM process of Inconel 625 alloys.

Key words: Additive manufacturing, Wire arc additive manufacturing, Cold metal transfer, Forming quality, Inconel 625

¹ **INTRODUCTION**

² WAAM technology is a 3D printing technology that 3 uses a metal welding wire as the input material $1-3$ $1-3$.

- ⁴ The arc energy sources used in the WAAM process
- ⁵ can include gas metal arc welding (GMAW or MIG),
- ⁶ cold metal transfer (CMT), gas tungsten arc welding
- σ (GTAW or TIG), and plasma arc welding (PAW)^{[4](#page-11-2)}.
- ⁸ Compared with other additive manufacturing (AM)

 technologies that use laser or electron beam sources and metal powder (e.g., from 0.1 to 0.6 kg/h), WAAM has a superior material deposition rate (from 3 to 8 $\frac{1}{2}$ kg/h)^{[5,](#page-11-3)[6](#page-11-4)}. Moreover, requirements for the environ- ment are not too strict, making this technology capa- ble of manufacturing large parts. The WAAM system 15 also has lower investmentcosts^{[7](#page-11-5)}. Currently, WAAM plays a very important role in many fields, such as the aerospace, construction, structural, nuclear energy,

 $_{18}$ and marine industries $_{8-13}$ $_{8-13}$ $_{8-13}$.

Many studies on the WAAM process for various alloys (e.g., steels, aluminum, titanium, and nickel- ²⁰ based alloys) have been reported in the literature. ²¹ These studies have focused mostly on the effects of 22 WAAM parameters on microstructures and mechan-
23 ical properties $14-17$ $14-17$. However, research evaluating 24 the influence of process parameters on the geomet-

25 ric shape and surface roughness of a product is lim- ²⁶ ited to only certain types of materials, such as stainless 27 steel and low-carbon steel. For example, Dinovitzer ²⁸ et al. 18 reported that when the welding speed increased, the surface roughness increased, and an in- ³⁰ verse relationship was observed between the current 31 applied during the WAAM process of the HASTEL- 32 LOY X alloy. Xiong et al. 19 19 19 provided a quantitative 33 evaluation approach for the surface roughness of thin- ³⁴ walled parts.

Inconel 625 is known as a nickel-based superalloy ³⁶ with high strength and good heat, corrosion, and 37

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 oxidation resistance. This alloy has a high bearing capacity and a wide temperature range, from cryo-⁴⁰ genic temperatures to 982[°]C ^{[14](#page-11-8),[15](#page-12-3)}. Several stud- ies have been published on the microstructures, me- chanical properties, corrosion resistance, residual stress, and defects of WAAMed Inconel 625 parts. Xu et al. 20 investigated the microstructures and mechanical properties of Inconel 625 parts fabri- cated by PAW-WAAMed with a continuous deposi- tion strategy (CDS) and an interpass cooling strategy (ICS).They reported that the fabricated parts revealed columnar dendrite structures decorated with a large amount of Laves phase (MC carbides and Ni3Nb), and ICS improved the mechanical properties and surface quality compared with those of CDS. Cheepu et al. 21 examined the effects of different deposition strategies for Inconel 625 alloys via super-TIG-WAAM.They re- ported that multi-pass beads with stringer and zigzag layering strategies could refine microstructures. Re- cently, Kumar et al. 22 22 22 performed a parametric study and optimization of weld beads in GMAW-WAAM of Inconel 625 via the RSM and DA methods. Motwani ω et al. 23 23 23 presented a study on multi-response opti- mization in CMT-WAAM of Inconel 625 via entropy weightage-assisted gray-based Taguchi analysis. Until recently, the influence of process parameters on the geometric characteristics of Inconel 625 products built via WAAM has rarely been reported. Therefore, this work aims to analyze the geometric character- istics of Inconel 625 thin-walled parts produced via CMT-WAAM technology and examine the effects of the welding speed (*v*) and linear heat input (*LHI*) on the part-forming quality. *LHI* refers to the amount of heat applied per unit length of weld track. In WAAM, *LHI* is the parameter with the greatest impact on the cooling rate that drives the shaping quality, mi- crostructures, and mechanical properties, whereas *v* is strongly related to the part geometry and produc- tivity. In the present study, we aimed to evaluate the geometric characteristics of the as-built part, includ- ing the average layer height (*ALH*), total width (*TW*), effective width (*EW*), and surface roughness of the thin-walled parts and the deposition efficiency (*DE*), and the effects of *v* and *LHI* on these characteristics. The outcomes enable us to discuss the actual process parameter selection for manufacturing Inconel 625 components with CMT-WAAM technology. The structure of this article is as follows: The materi- als and experimental procedures are described in sec-87 tion 2. The primary findings regarding the geomet- ric properties and surface roughness of the building materials are provided in the Results and Discussion section. The last section, conclusions, provides an

overview of significant findings and suggestions for the future. $\qquad \qquad \text{92}$

MATERIALS AND METHODS

Materials

In this research, DAIKO SF 625 welding wire, which has a broad operating temperature range from -269[°]C to 1000[°]C, was used. Its diameter is 1.2 mm, and the chemical composition of the wire is presented in Ta-ble [1](#page-2-0) (according to the AWS $A5.14$ standard):

The material used in the experimental process was 100 S45C (JIS G4051) steel. The chemical composition of 101 the substrate included 0.48 C, 0.35 Si, 0.9 Mn, 0.03 P, ¹⁰² 0.035 S, 0.2 Ni, 0.2 Cr, and 0.3 Cu (in wt.%). Its di- ¹⁰³ mensions are 200 mm \times 200 mm \times 10 mm in length, 104 width, and height, respectively. On the other hand, the plate was cleaned before the experiment.

CMT-WAAM system 107

All the samples were fabricated on the CMT-WAAM 108 system, as shown in Figure [1.](#page-2-1) It consists of a CMT 109 welding unit (TPS 320i) and a 6-axis robot (FD-V8 ¹¹⁰ OTC Daihen). During welding, the motion of the ¹¹¹ welding wire is regulated with an average frequency 112 of approximately 70 Hz $^{24-27}$ $^{24-27}$ $^{24-27}$ $^{24-27}$ $^{24-27}$, and the power source is 113 supplied intermittently during each short-circuit pe- 114 riod to create the arc and melt the metal wire. As a ¹¹⁵ result, molten metal droplets were steadily controlled ¹¹⁶ into the welding pool. 117

With the CMT welding principle, the welding wire is 118 first moved closer to the substrate to start the welding 119 process. An arc is formed, and the welding head is ¹²⁰ controlled to process the molten metal into the weld- ¹²¹ ing pool by moving the welding wire closer and far- ¹²² ther away to minimize the thermal effect in the weld- ¹²³ ing area. At the end of a welding line, the process is ¹²⁴ repeated. The weld beads are built on each other until 125 the part is completed.

Building thin-walled samples and data col- ¹²⁷ **lection methods** 128

In this study, three rectangular thin-walled samples 129 were subjected to three different welding speeds (*v* ¹³⁰ $= 65, 50,$ and 35 cm/min), and the wire feed speed 131 (*WFS*) was held constant at 6.5 m/min, as shown in ¹³² Table [2.](#page-3-0) The linear heat input (*LHI*) is calculated ac- 133 cording to Eq. (1) .

$$
LHI = \eta \times \frac{U \times I}{v} (J/mm)
$$

where η is the energy conversion efficiency (normally, 135 η = 85% for CMT), and U and I are the voltage and 136

¹³⁷ current, respectively. With the CMT power source, ¹³⁸ U and I were automatically set according to the value of the wire feed speed, as shown in Table [2](#page-3-0).

 The thin-walled samples were made of 20 layers, and the programmed dimension of each layer was the same, with dimensions of 130 mm in length and 60 mm in width. The starting point, the ending point, and the transfer point between the layers are the same and are in the middle of the length dimension. Dur- ing the deposition process, the arc was emitted con- tinuously according to the CMT principle. Moreover, commercially pure argon was used (99.99% Ar) with 149 a flow rate of 16 L.min⁻¹ to protect the welding pool. All the samples were fabricated at room temperature. The distance between the torch tip and the workpiece surface was fixed at 10 mm. The angle between the axis of the welding torch and the substrate was also maintained at 90 degrees.

¹⁵⁵ After fabrication, all the samples were scanned by a

¹⁵⁶ Kreon Zypher II scanner to collect their actual shape.

157 The scanned data were then processed with Geomagic

¹⁵⁸ Design X software, as shown in Figure [2](#page-3-1).

¹⁵⁹ In this study, the geometric characteristics of the thin

- ¹⁶⁰ walls investigated are illustrated in Figure [3](#page-5-0), including
- ¹⁶¹ the total width (*TW*), the effective width (*EW*), the
- ¹⁶² total height (*TH*), and the effective wall height (*EH*). ¹⁶³ The average layer height (*ALH*) is determined by the

¹⁶⁴ ratio between *TH* and the number of printed layers.

The method used to analyze the geometrical charac- 165 teristics is presented in Figure [4](#page-6-0). Each thin-walled ¹⁶⁶ sample was evaluated at four cross-sections (cs1, cs2, 167) cs3, and cs4). The distance between two adjacent ¹⁶⁸ cross-sections is 25 mm. cs1 and cs2 are symmetri- ¹⁶⁹ cal with respect to cs4 and cs3, respectively, through 170 the center plane along the length of the scanned sam- ¹⁷¹ ple. This ensures the survey along the entire length 172 of the deposition path and monitoring the difference ¹⁷³ in profile at the beginning and end of the deposition ¹⁷⁴ line. These intersection profiles were processed in Au- ¹⁷⁵ toCAD software to measure *ALH*, *TW*, *EW*, and *DE*. ¹⁷⁶ The material deposition efficiency (*DE*) is a value that 177 represents the efficiency of the entire WAAM process ¹⁷⁸ in general. It is determined by the ratio of the effective 179 cross-sectional area to the total cross-sectional area at ¹⁸⁰ the local location under consideration, as described 181 in Eq. (2), where the effective cross-sectional area is 182 calculated via the values of effective width (*EW*) and ¹⁸³ effective height (*EH*).

$$
DE = \frac{EH \times EW}{Total\ area}
$$
 (2)

The process for evaluating the surface roughness pa- 185 rameters is shown in Figure [5](#page-6-1). For each sample, two ¹⁸⁶ portions of the opposite surface on each wall along 187 the length of the sample were cut. Each portion has 188 the same dimensions of 70 mm \times 30 mm. These four 189

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Sample No. 1

Figure 2: 3D scanned data of the samples

 surface samples were subsequently used to analyze the surface roughness parameters via Omnisurf 3D soft- ware. The studied roughness parameters are the max-193 imum roughness S_z , average roughness S_a , and mean 194 square roughness S_q , which are calculated via Eqs. (3), (4), and (5), respectively.

$$
S_z = z_{max}(x, y) - z_{min}(x, y)
$$
 (3)

$$
S_a = \frac{1}{S} \int \int_S |z(x, y)| dx dy
$$
\n
$$
S_q = \sqrt{\frac{1}{S} \int \int_S z^2(x, y) dx dy}
$$
\n(4)

196 where $z(x, y)$ represents the coordinates of the grid ¹⁹⁷ points and represents the surface area.

RESULTS ¹⁹⁸

Geometric characteristics 199

The results regarding the geometric characteristics 200 (including *TW*, *TH*, *EW*, *EH*, and *ALH*) of the thin- ²⁰¹ wall samples fabricated via CMT-WAAM are dis- ²⁰² played in Tables [3](#page-4-0) and [4,](#page-5-1) respectively. For each type ²⁰³ of output result, the average values from the measure- ²⁰⁴ ments were calculated to construct graphs. 205

The graphs that depict the influence of the welding 206 speed *v* on *ALH*, *TW*, and *EW* are presented in Fig- ²⁰⁷

ure [6](#page-8-0). The *ALH* decreases proportionally with the 208 welding speed *v*. The *ALH* decreases from 2.63 mm ²⁰⁹ to 1.83 mm when ν increases from 35 cm/min to 210 65 cm/min. *ALH* tends to decrease rapidly at lower ²¹¹

Table 4: Average height value of each layer (ALH)

²¹² speeds (from 35 to 50 cm/min), whereas it slowly de-213 creases at high values of v , from 50 to 65 cm/min (Fig-²¹⁴ ure [6](#page-8-0)a).

²¹⁵ Figure [6b](#page-8-0) shows that both the values of *TW* and *EW*

- ²¹⁶ gradually decrease with increasing *v*. Specifically, the
- ²¹⁷ overall width of the wall (*TW*) is 9.39 mm, 8.00 mm,
- and 6.83 mm when ν is set at 35, 50, and 65 cm/min,
- ²¹⁹ respectively. The effective width of each sample (*EW*) ²²⁰ is also proportional to the overall thickness, and its
- 221 values are 5.9 mm 5.04 mm and 4.3 mm at $v = (35, 50, 50)$
- ²²² and 65) cm/min, respectively.

²²³ Figure [7](#page-8-1) illustrates the four surface portions of the ²²⁴ thin-walled sample "No. 1" that were extracted from

- ²²⁵ the 3D scanned data through the procedure in Figure
- ²²⁶ 5. These representative surfaces were used to measure
- 227 surface roughness parameters $(S_z, S_a, \text{ and } S_a)$ via Om-
- ²²⁸ nisurf 3D software. Figure [8](#page-8-2) shows the morphology
- ²²⁹ characterization of the surface portion in Figure [7](#page-8-1).

²³⁰ The surface roughness parameters of all the samples

²³¹ are shown in Table [5](#page-9-0) and Figure [9.](#page-9-1) The average mea-²³² surement values are represented to evaluate and com-

²³³ pare all the samples.

The maximum roughness S_z value decreases from 168.968 µm to 108.464 µm when *v* increases from 35 to 50 cm/min, respectively. When *v* increases from 50 to 65 cm/min, *Sz* tends to increase from 108.464 μ m to 128.811 μ m. Similarly, the average rough-239 ness S_a and the mean square roughness S_a have simi-240 lar trends. For example, S_a decreases from 5.902 μ m ²⁴¹ to 4.326 μ m when *v* increases from 35 cm/min to 50 cm/min, whereas it increases from 4.326 μ m to 5.394 μ m *as v increases* from 50 cm/min to 65 cm/min. S_q decreases from 9.005 µm to 6.544 µm when *v* in- creases from 35 cm/min to 50 cm/min, and it in- creases from 6.544 µm to 8.115 µm *as v increases*from 50 cm/min to 65 cm/min.

²⁴⁸ **DISCUSSION**

²⁴⁹ **Effects of** *LHI* **on** *ALH***,** *TW***, and** *EW*

 As mentioned previously, the linear heat input (*LHI*) plays a crucial role in determining the dimensions and properties of the deposited layers. When the *LHI* increases, a greater amount of energy is applied to melt the filler wire and the base metal, resulting in larger melting pool dimensions and deeper penetra- tion into the base material. Consequently, increas- ing *LHI* tends to result in thicker layers and increased *ALH* (Figure [11a](#page-10-0)). Similarly, *TW* and *EW* increase with increasing *LHI* (Figure [11b](#page-10-0)).

²⁶⁰ The effect trend of *LHI* on*ALH*, *TW*, and *EW* is oppo-²⁶¹ site to that of *v*. Figure [6](#page-8-0) shows that all the *ALH*, *TW,* ²⁶² and *EW* values decrease with increasing v. This can be explained by the nature of the process parameters. v_{263} indicates the rate at which the amount of molten ma- ²⁶⁴ terial is deposited into a welding line, whereas *LHI* ²⁶⁵ refers to the amount of heat applied per unit length ²⁶⁶ of deposition. As indicated in Eq. (1), the *LHI* is high ²⁶⁷ at a low value of $v^{28,29}$ $v^{28,29}$ $v^{28,29}$ $v^{28,29}$ $v^{28,29}$. At a fixed value of *WFS*, when 268 the *LHI* increases (or *v* decreases), the ability to melt 269 the metal wire is greater, more material is melted and 270 added to the weld pool, and the solidification process 271 also takes longer 19 . Therefore, the melting pool and 272 weld bead become increasingly larger. This is why all 273 the characteristics *ALH*, *TW*, and *EW* increase (Fig- ²⁷⁴ $ure 11)$ $ure 11)$ $ure 11)$. 275

Effects of *LHI* **on surface roughness** ²⁷⁶

The influence of *LHI* on surface roughness parameters 277 is similar to the influence of ν , meaning that they de- 278 crease when the *LHI* increases from 191 to 245 J/mm 279 and increase again when the *LHI* increases from 245 280 to 352 J/mm (Figure [12\)](#page-11-9). The reason is that when 281 *WFS is maintained*, increasing *v* reduces the *WFS*/*v* ²⁸² ratio, and the staircase effect between layers decreases, ²⁸³ leading to a decrease in surface roughness, and the ²⁸⁴ surface quality gradually stabilizes. When ν is too 285 low, the amount of heat in the weld pool is large, and ²⁸⁶ the cooling rate is low. After deposition, the sur- ²⁸⁷ face unevenness and surface roughness are high. As 288 *v* increases, this heat buildup decreases, the material ²⁸⁹ melts and solidifies more stably, the surface roughness 290 becomes more uniform, and the values decrease ^{[19](#page-12-2)}. 291 However, if *v* continues to increase, the stability of the 292 arc gradually deteriorates. This adversely affects the ²⁹³ surface roughness. As a result, the surface roughness 294 increases, and the surface quality deteriorates. 295

CONCLUSIONS ²⁹⁶

This study investigated the influence of various pro- ²⁹⁷ cess parameters, specifically the welding speed ν and 298 linear heat input *LHI*, on the geometric characteris- ²⁹⁹ tics and material deposition efficiency of the WAAM 300 process for Inconel 625 alloys. The main conclusions 301 are expressed as follows: 302

• As the welding speed *v* increases, the *LHI* and the ³⁰³ volume of material added to the melting pool de- ³⁰⁴ crease, causing the average layer height, total width, ³⁰⁵ and effective width to decrease.

• The average layer height, total width, and effective 307 width increase when the *LHI* increases. 308

• With increasing welding speed *v* and *LHI*, the max- ³⁰⁹ imum roughness S_z , average roughness S_a , and mean 310 square roughness S_q decrease. They continue to in- 311 crease as the welding speed ν or *LHI* continues to in- 312 crease. 313

Figure 7: The surface portions of sample No. 1

Sample	Number of measurements	S_z (μm)	\mathbf{S}_a (μm)	S_q (μm)
No. 1	$\mathbf{1}$	96.05	5.29	7.76
	$\overline{2}$	129.18	5.41	8.20
	\mathfrak{Z}	156.06	5.04	7.81
	$\overline{4}$	133.96	5.84	8.70
	Average	128.811	5.394	8.115
	Standard deviation	± 24.783	± 0.336	± 0.434
No. 2	$\mathbf{1}$	124.912	4.019	6.106
	$\overline{2}$	122.316	4.257	6.586
	3	89.041	4.285	6.319
	$\overline{4}$	97.586	4.741	7.166
	Average	108.464	4.326	6.544
	Standard deviation	± 17.870	\pm 0.302	0.459
No. 3	$\mathbf{1}$	194.657	6.235	9.6
	$\overline{2}$	169.984	5.775	8.565
	$\overline{\mathbf{3}}$	157.580	5.562	8.792
	$\overline{4}$	153.649	6.034	9.064
	Average	168.968	5.902	9.005
	Standard deviation	18.487	0.294	0.446

Table 5: Surface roughness parameters measured via Omnisurf 3D software

10

 The findings of this study provide insight into the ef- fects of process parameters (*v* and *LHI*) on the geo- metric characteristics of as-built parts, which can sup- port the adjustment of the process parameters in the CMT-WAAM process of Inconel 625 alloy to achieve the expected quality. Additionally, the estimated *Sz* value can be considered a machining allowance for finishing operations. In this study, only the effects of the process parame-

 ters on the geometric characteristics of the walls were observed. In future works, it will be interesting to de- velop regression models for all the characteristics with high accuracy. These models can be used to predict the proper parameters and more process parameters to achieve the desired quality. Moreover, it is also important to investigate the microstructures and me- chanical properties of the as-built material to confirm its feasibility in real applications.

LIST OF ABBREVIATIONS

Figure 12: Influence of *LHI* on *S^z* , *Sa*, and *S^q*

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- ³⁵⁰ Foundation for Science and Technology Development
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³⁵² **DATA AVILABILITY**

³⁵³ All the data generated or analyzed during this study ³⁵⁴ are included in this published article.

³⁵⁵ **CONFLICT OF INTEREST**

³⁵⁶ The authors declare that they have no competing in-³⁵⁷ terests.

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